Structure & Properties Of Electroless Ni-P-B₄C Composite Coatings

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The surface image and inner structure of electroless Ni-P-B₄C deposit layers were observed and analyzed by SEM, optical microscope and X-ray diffractometer. The results showed that with increase of the amount of $B_{4}C$ particles in the plating solution, the number of the surface nodular protrusions increased. The particles of B₄C evenly dispersed on the Ni-P matrix and adhered well in the coatings. The presence of the B₄C particles does not change the phase composition. The mechanical properties of composite coatings were measured. The results indicated that with increase of the content of B_AC , the hardness and wear resistance were obviously improved, but the critical load was decreased.

It is well known that there are three types of material failure: corrosion, wear and fatigue. Wear is the main failure mode for many machine components. Improving surface hardness is one method to upgrade the wear resistance of materials. At present, to improve surface hardness and wear resistance, many treatment methods are adopted; for example, electroplating, carbon case hardening, nitrogen hardening, PVD, CVD, ion implantation, etc. Every technology has its limitations, however. Some have poor feasibility, and some are expensive. Electroless Ni-P composite plating is one of the new technical metal surface treatments. It offers excellent technical feasibility (good throwing power, uniform coating thickness, good surface smoothness, etc.), as well as other functional properties.

Many reports have been published worldwide that are mainly focused on electroless Ni-P composite plating.1-7 Although there are a few reports concerning Ni-P-B₄C deposits, little data was given. In this paper, the authors emphasize study of the structure and mechanical properties of Ni- $P-B_4C$ deposit coatings.

Experimental Procedure

Electroless Composite Plating Solution

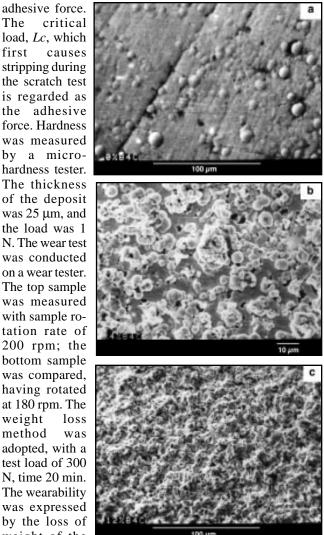
In the plating solution, sodium butanate and amino acetic acid are used as complexing agents, with sodium hypophosphite as reducing agent. Solution pH, 4.5 to 5.0, temp 85 °C. The particle size of the B₄C is W5. Mechanical agitation is employed at the rate of 750 rpm.

Pre-cleaned Substrates of Gray Cast Iron

The preparation and plating sequence for the substrates was chemical de-oiling and rust removal -> washing with deionized water —> chemical activation (15% H_2SO_4 solution), 25 °C, 8 sec) \longrightarrow pre-plating Ni \longrightarrow electroless plating.

Assessment of Coatings

The surface images before and after the wear test were observed by SEM, and the phase was analyzed by X-ray diffractometer. A scratch test was adopted to measure the first causes stripping during the scratch test is regarded as the adhesive force. Hardness was measured by a microhardness tester. The thickness of the deposit was 25 µm, and the load was 1 N. The wear test was conducted on a wear tester. The top sample was measured with sample rotation rate of 200 rpm; the bottom sample was compared, having rotated at 180 rpm. The weight loss method was adopted, with a test load of 300 N, time 20 min. The wearability was expressed by the loss of weight of the test



sample before Fig. 1—SEM micrographs of composite coatings: (a) and after the Ni-P; (b) Ni-P-19%B₄C; (c) Ni-P-32%B₄C.

Results & Discussion

Surface Image & Inner Structure

Figures 1(a)(b)(c) are SEM micrographs of Ni-P, Ni-P-19% B₄C and Ni-P-32% B₄C composite coatings, respectively. It can be seen from the figures that there are a few nodular protrusions and pores in the surface of the Ni-P deposit; more in the surface of the Ni-P-19%B₄C coating, but the surface of the Ni-P-32% B₄C deposit is almost covered with nodular protrusions. The micrographs indicate that, with increase of the content of $B_{A}C$ particles, the surface nodular protrusions increase, but their size gets smaller. In addition, they show that the B₄C particles are partly exposed or wholly embed-

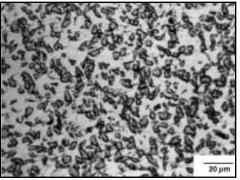


Fig. 2—Optical micrograph of composite coating.

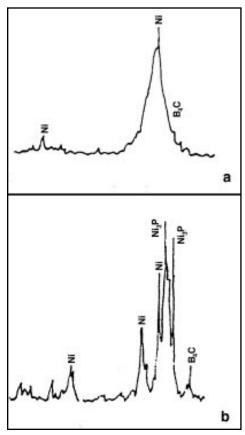


Fig. 3—X-ray diffraction curve of Ni-P-26%B₄C coatings: (a) as deposited; (b) after heat treatment for one hr at 350 °C.

shows that there are B_4C particles in the amorphous Ni-P matrix. Figure 3b shows the X-ray diffraction curve of Ni-P-26% B_4C plating layer after heat treatment for one hr at 350 °C. The curve shows that the coating consists of Ni, Ni₃P and B_4C .

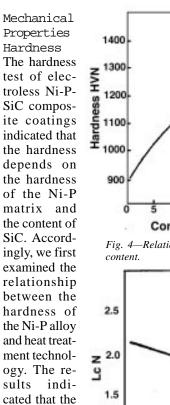
Figure 3c reveals that the B_4C content in the plating layer affected the diffraction intensity of Ni and Ni₃P. The increase of the B_4C particle content caused a decrease of the Ni and Ni₃P diffraction intensity attributable to the change in quantities of the constituents.

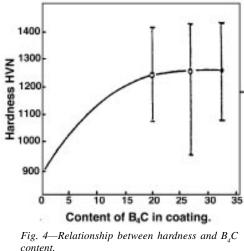
In the X-ray diffraction curve of the Ni-P deposit,⁷ it can be seen that the B_4C particles only strengthen the intensity of diffraction. From these results, it is clear that the presence of B_4C particles doesn't change the structure and phase composition of the Ni-P matrix.

ded in the Ni-P matrix. Figure 2 is an optical micrograph of the surface-polished Ni-P-26%B₄C composite coating. It can be seen that B₄C particles are evenly dispersed in the coatings.

The amount of particles in coatings depends on the amount of the particles added to the plating solution, the agitation rate and particle diameter. After the latter two factors were determined in the experiment, it was found that the amount of B₄C particles in the coatings increases with increase of particles in the plating solution.

Figure 3a shows the X-ray diffraction curve of Ni-P- $26\% B_4C$ in the deposit. The curve





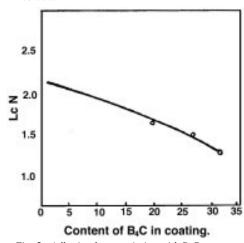


Fig. 5—Adhesive force variation with B_4C content.

Figure 4 shows the relationship between B_4C content and the hardness of Ni-P-B₄C at one hr heat treatment at 350 °C. The results show that the hardness increased considerably when the coatings contained B_4C particles, but the test data have some scattering, which may be because the load is so small that the indentation can only impinge partially on B_4C particle, or wholly on the Ni-P matrix. The real hardness of composite coatings cannot, therefore, be obtained, but it is nevertheless true that with increase of the B_4C content in the deposit coatings, the entire hardness increased.

Adhesive Force

greatest hard-

ness of the

Ni-P deposit

coatings is

HV 912 after

one hr heat

treatment at

350 °C.

A scratching test was adopted to measure the adhesive force of composite coatings. The results are shown in Fig. 5. The curve indicates that the higher the content of B_4C , the poorer the adhesive force.

Figure 6 shows the micrograph of the spot where the coatings strip during scratching. With the load on the Ni-P layer increasing (direction of arrow), it first appeared as a crack, then stripped large lumps, but on the Ni-P-32%B₄C coatings, only small lumps were stripped. It indicates that adding B₄C particles destroys the continuity of the Ni-P matrix and that *Lc* cannot be used to evaluate exactly the real adhesive force of composite coatings. Obviously, the force needed to cause stripping of large lumps is larger than the force to cause stripping of small lumps. Indentation and file tests qualitatively indicated that the Ni-P-B₄C composite coating has excellent adhesion to cast substrates.